



SUBSTITUTE SPECIFICATION



PARTICULATE REINFORCED ALUMINUM COMPOSITES,
THEIR COMPONENTS AND
THE NEAR NET SHAPE FORMING PROCESS OF THE COMPONENTS

5 Field of the Invention

This invention involves particulate reinforced Aluminium (Al) based composites, their components and the near net shape forming process of the components.

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Background

Particulate reinforced Al-based composites offer higher specific stiffness and specific strength, good wear-resistance, better fatigue durability, lower thermal expansion coefficient and good dimensional stability, as compared to conventional Al alloys. In addition, the formulas of the composites can be designed in a wide range to meet specific properties, while the conventional Al alloys do not have such capabilities of designability. Therefore, many countries have spent substantial amounts of investments to develop this type of composites, and some of them have been successfully applied in aerospace, military and civilian industries. With the increasing demand for particulate reinforced Al-based composites, fabrication of high performance and cost effective composite components are the focus of current research and development activities. High performance refers to higher mechanical and physical properties with good machinability; while cost effective means to minimize the cost of the composite billets and their final component-forming process cost, especially for the complex components.

At present, four major fabricating processes are used to make particulate reinforced Al-based composite billets, including powder metallurgy (PM), agitation casting, spray forming and squeeze casting. Two key issues to be resolved in the current composite billet fabrication processes are: 1) improvement of the uniformity of the reinforced particle distribution and 2) enhancing their bonding strength in the Al matrix.

It is well-documented that composite properties are controlled by the following key parameters, such as reinforced particle size, their uniformity of distribution, and their interfacial bonding with the matrix. Also, the composite

5 machinability strongly depends on the particle size. Small reinforced particle composites have better machinability. With a view to fabricating high performance and cost effective composites, the desirable parameters include small reinforced particles and uniform distribution, and good interfacial bonding in the matrix. Among the four fabricating processes indicated above,

10 the powder metallurgy (PM) process represents the best one to meet the above parameters. Nevertheless, due to a large particle-size ratio ($\geq 11 - 28$) between the raw Al powder particles ($40 - 100\mu\text{m}$) and the reinforced particles ($\leq 3.5\mu\text{m}$), it is difficult to achieve a uniform distribution of the reinforced particles in the matrix using the conventional mechanical mixing

15 processes. In addition, the surface oxide layer of the Al powder will deteriorate the interfacial bonding strength with the matrix. Thus, high-quality and easy machining composites are hardly fabricated through the ordinary mechanical mixing processes.

20 US patent number US 3,591,362 by *Benjamin et. al.* provides a theoretical approach to solve this problem . Using a high-energy ball-milling technique, the Al alloy matrix powder particles are deformed repeatedly under grinding and impact by high-energy balls, and a cold-weld layer forms on the ball surface. This cold-weld layer will fall off and be crushed by the continuous

25 work-hardening. Finally, fine composite powers are obtained. Later, US patent number 3,740,210 invented a raw material for the dispersion-strengthened Al composite, consisting of Al powder and its oxide powder. In this process, the raw powders with surfactant are dry-grounded. However, the properties of the composite billet made are deteriorated because the fine

30 composite powders contain the surfactant. Another US patent (number 4,946,500) introduced a method to fabricate Al-based composites, consisting of Al-alloy powder and reinforced particle powder. The raw powders are mixed under a high-energy ball-milling process without adding any surfactant, thus

eliminating the negative effect on the properties of the final composite billet. However, cold-welding tends to be more severe during ball-milling without adding surfactant, resulting in an unstable mixing / homogenizing process. Thus, it is not suitable for continuous industrial production. This patent does not specify how to solve the cold-weld problem in case no surfactant is added. In addition, the particle size of the ball-grinding composite powder is too large when surfactant is absent during the milling process. As a result, the billet produced cannot meet requirements in the subsequent compressing forming process.

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After acquiring the high performance composite billet fabrication technology, the next key issue is how to reduce the process cost, especially for the complex shape composite components. The near net shape approach is the most cost effective way for making composite components. Machining and mold-forging are the commonly used fabricating methods for components. However, machining would increase the cost due to the composite's poor machinability. Cost of mold forging is also higher because of the composite's poor plastic deformation characteristics.

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Semisolid forming components is one of the near net shape forming processes. Thanks to the fine particle size of either the Al matrix powder or the reinforced particles, the composite billet should have a thixotropic characteristic for semisolid processing. A semisolid near net shape forming composite approach has been proven successful using a spray forming composite billet as described in a recent US patent (US patent number 6,135,195). This patent invented a thixotropic composite of SiC/2xxxAl, the composite billet being made by a spray forming process. To ensure the thixotropy of the composite billet, additional Si of 1 - 5 wt % is added into the standard Al alloy. Also, a well-controlled double heating method is used.

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However, this patent does not state whether or not this material has its thixotropic nature without adding more Si. Normally, the composite billet prepared by the spray forming process exhibits poor macroscopic distribution of the reinforced particles, and the composite properties are, therefore, not consistent.

Summary of the Invention

5 This invention seeks to develop a high-performance, low-cost particulate reinforced Al-based composite and a high-performance, particulate reinforced Al-based composite billet that is easily machinable.

10 Another object of the present invention is to develop a near net shape forming process for the particulate reinforced Al-base composite components by employing high-energy powder mixing techniques to produce high-performance, particulate reinforced Al-based composite billets that are easily machinable, then near net shape forming components using a semisolid forming technique, and finally, producing high performance and cost effective near net shape forming composite components for industrial applications.

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According to a first aspect of the invention, there is provided a type of particulate reinforced Al-based composite which comprises reinforced particles and aluminium alloy, wherein: (1) the reinforced particles are dispersively and uniformly distributed in an aluminum alloy matrix, and forms
20 interfacial bonding with the matrix; (2) the average particle size of the reinforced particles is 0.1 - 3.5 μ m; and (3) the volume percentage of the reinforced particles is 10 - 40%.

25 In the preferred embodiment of the invention, the reinforced particulates have high hardness, high elastic modulus and strength and low density. The reinforced particulates may be one of the following: B₄C (boric carbide), SiC (silicon carbide), Al₂O₃ (alumina) and AlN (aluminum nitride). The Al matrix alloys may be any of the Al alloys, including one of the following: 2xxx, 6xxx and 7xxx.

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According to a second aspect of the invention, there is provided a particulate reinforced Al-based composite component, wherein the component is made from a composite billet of the said particulate reinforced Al-based composite.

According to a third aspect of the invention, there is provided a method of forming a type of particulate reinforced Al-based composite comprising the steps of (1) according to a desired volume percentage of reinforced particles in an Al-based composite, determining a weight percentage of the required reinforced particles; (2) based on the required weight percentage of reinforced particles in the composite, determining a required weight of the reinforced particle and corresponding weight of an aluminum alloy powder; (3) loading required amounts of reinforced particles, Al-based alloy powder and steel balls into a balling drum of a high-energy ball-mill, then carrying out high-energy ball-milling to form a composite powder; (4) adding liquid surfactant, and continuing with ball-milling; (5) molding the composite powder into a desired shape through cold isostatic pressing; (6) processing the cold isostatic pressed shape into a compact billet by means of vacuum sintering or vacuum hot-pressing; then (7) heating the compact billet, and undertaking semisolid die-cast forming to produce a near net shape composite component.

It will be convenient to hereinafter describe the invention in greater detail by reference to the accompanying drawings, which illustrate one embodiment of the invention. The particularity of the drawings and the related description is not to be understood as superseding the generality of the broad identification of the invention as defined by the claims.

Brief Description of the Drawings

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Figure 1 shows the optical metallographic structure ($\times 200$) of Composite AlN/6061 after a 6-hour high-energy ball-milling (AlN powder and 6061 alloy powder);

Figure 2a shows the particle distribution of B_4C after common mechanical mixing;

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Figure 2b shows the particle distribution of B_4C after high-energy ball-milling;

Figure 3 shows the cold-welding stripes in the B_4C /6061 composite powder;

Figure 4 shows a vacuum hot-pressed billet made from composites of 35vol% AlNp/6061Al and 35vol%SiCp/6061Al; and

Figure 5 shows the optical microcosmic structure ($\times 500$) of a thixotropic composite SiCp/A1.

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Detailed Description of the Preferred Embodiment

The preferred embodiment of a method of producing the components of particulate reinforced Al-based composites according to the invention involves the following steps: 1) A volume percentage of 10 - 40% of the reinforced particles (an equivalent weight percentage of 9.3 - 50.9%) in the composite is determined; then according to the weight percentage of reinforced particles, the weight of the required reinforced particles and the balanced raw Al-base alloy powder can be calculated; 2) load the required volume of reinforced particles, Al-alloy powder required and the steel balls into a balling drum, and variable-speed high-energy ball-milling undertaken from 1 - 10-hours. The weight ratio between the steel balls and the powder is 10 — 50 : 1; the rotational speed of the high-energy ball-mill is arranged into two stages: first a low speed stage, then a high-speed stage. At the low-speed ball-milling stage, the rotational speed is 100 — 150rpm, milling for 10 — 40 minutes; while at the high-speed ball-milling stage, the speed is 150 — 300rpm, milling for 20 — 600 minutes; 3) adding a liquid surfactant and ball-milling for 0.5 — 2 hours in a temperature range of 15 - 80°C. The weight ratio between the balls and the powder is 10 — 50: 1, and the rotational speed of the high-energy ball-milling is 100 — 300 rpm. The composite powders are produced through steps 1) to 3); 4) the composite powders are further subjected to a cold isostatic pressing to form a green billet under a pressure of 200 - 1000 MPa for 1 - 10 minutes. The green billet is 70 - 80% of the theoretical density; and 5) hot compressing the green billet into a dense billet by a vacuum-sintering or a hot-forming process at a temperature of 450 - 600 °C, under a pressure of 36 - 70 MPa and a vacuum of no less than 1.5×10^{-2} Pa; and 6) double heating the dense billet at 600 - 660°C, and when reaching a liquid-

phase content of 60 - 70%, undertaking semisolid squeeze or die casting for near net shape forming the composite components.

At step 2, high-carbon steel balls with a diameter of $\phi 5 - \phi 8\text{mm}$ are considered optimal. The average particle size ratio between the reinforced particles and the Al-based alloy powder can be selected within a wide range. The average size of the reinforced particles is greater than $0.1\mu\text{m}$ and the average size of the Al-based alloy powder is larger than $10\mu\text{m}$. The particle size ratio (reinforced/matrix) can be selected within the following range: $0.1 - 100\mu\text{m}/10 - 210\mu\text{m}$.

In the component fabricating process, the designed 10 - 40% volume percentage content (9.3 - 50.9% weight percentage) of the reinforced particles, and the balanced Al matrix powder is prepared. The reinforced particles with an average particle size of $0.1 - 100\mu\text{m}$ and the Al alloy powders of $10 - 210\mu\text{m}$ will be used; mixing the powders with $\phi 5 - \phi 8\text{mm}$ high-carbon steel balls and loading them into a balling drum, under a $0.1 - 10\text{Pa}$ vacuum, followed by filling an inert gas (preferably nitrogen or argon). The pressure of the filled nitrogen or argon is $1.01 \times 10^5\text{Pa} - 1.1 \times 10^5\text{Pa}$. The balling drum is water-cooled to 25°C , starting variable-speed high-energy ball-grinding for 1 - 10 hours. The rotation speed of the high-energy ball-mill is divided into a low-speed range of $100 - 150\text{rpm}$ for 10 - 40 minutes, and a high-speed of $150 - 300\text{rpm}$ for 20-600 minutes, respectively. The variable-speed high-energy ball-milling helps to avoid cold-welding without adding surfactant, and thus, to ensure a smooth milling process. After ball-milling, first add 10 - 50ml liquid surfactant, under a vacuum of $0.1 - 10\text{MPa}$, then fill with nitrogen or argon gas at a pressure of $1.01 \times 10^5\text{Pa} - 1.1 \times 10^5\text{Pa}$. In the condition of no water cooling, the mixed powders can be ground into $10 - 120\mu\text{m}$ after high energy balling about 0.5 to 2 hours at $15 - 80^\circ\text{C}$. Pack and seal the composite powder in a vacuum rubber package and subject it to cold isostatic pressing at a pressure of $200 - 1000\text{MPa}$ for 1 - 10 minutes. The formed green composite billet reaches 70 - 80% of its theoretical density. The green billet is hot compressed into a compact billet through a vacuum-

sintering or a hot-pressing at 450 - 600°C, under a pressing pressure of 36 - 70MPa and a vacuum of no less than 5×10^{-2} Pa. The compact billet is finally heated at 600 - 660°C in a specially-designed induction furnace, preferably by double-heating. When a liquid-phase content reaches about 60 - 70% in the matrix, it is ready for the semisolid squeeze casting process.

The volume of the added surfactant in the milling process is 10 - 50ml. The surfactant can be any of the following organic solvents, such as gasoline, aviation gasoline, methanol or ethanol. At step 3, the weight ratio between the steel balls and the total weight of reinforced particles and Al-alloy powder is 10 - 50: 1, and the rotational speed is 100 - 300rpm.

The average size of the reinforced particle is one of the key factors affecting the overall properties of the composite. In general, large-sized particles (>3.5µm) help to improve the elastic modulus and strength but reduce the plasticity significantly. On the contrary, small-sized particles (<3.5µm) and submicron particles are capable of maintaining a high plasticity and ductility, and also increase the elastic modulus and strength, which is favorable to the secondary process and machinability. The average reinforced particle size can be controlled within the range of 0.1 - 1µm. Fig. 1 shows the particulate reinforced composite with an average particle size of 1.5µm after 6 hours balling. The following parameters will affect the reinforced particle average size, including the weight ratio between the steel balls and two types of feed powders (the ball-powder ratio), the rotational speed, and the high-energy ball-milling time. The high performance composite powders can be obtained using the following parameters, including a larger ball-powder ratio, a 180 - 300rpm high rotation speed and a longer ball-milling time of 4 - 10 hours. In the ball-milling mixing process mentioned above, the ball-powder ratio can be chosen in a range of 10 - 50: 1, however, a range of 20 - 50: 1 is more preferable.

At Steps 2 and 3, the ball-powder ratio is the ratio between the weight of steel balls and the total weight of reinforced particles and the Al alloy powder, and selection of a specific ball-powder ratio depends on the requirements for the

average size of reinforced particle and the ball-milling time: the smaller the average size of reinforced particles, the greater the ball-powder ratio; the shorter the ball-milling time, the greater the ball-powder ratio. In steps 2 and 3, the rotation speed of the high-energy ball-mill can be chosen from 100 - 300rpm. A speed range of 180 - 300rpm is more preferable. In step 2, the rotation speed of high-energy ball-milling is 150 - 300 rpm, and a range of 180 - 300 rpm is more preferable. The specific rotation speed mostly depends on the requirements for the average size of reinforced particle and the ball-milling time: the smaller the average size of reinforced particle, the higher the rotation speed needed; the shorter the ball-milling time, the higher the rotation speed required. In addition, the rotation speed must be appropriate to prevent powder from adhering.

During the step 2 ball-milling process, a variable-speed high-energy ball-milling is implemented to prevent the adhesion of Al-based alloy powder. Initially, a low-speed high-energy ball-milling is used to achieve the work hardening of the Al-based alloy powder, then a high -speed high-energy ball-milling is adopted to make composite powders.

With the addition of surfactant, the composite power can be crushed rapidly, and to allow its average particle size to meet the requirements of the subsequent forming process, the particle size of the composite powder should be in the range of 10 - 120 μ m.

The uniform distribution of reinforced particles in the matrix is a major issue to be resolved in the preparation of composites. When using the commonly used mechanical mixing, the reinforced particle distribution uniformity is mainly controlled by the physical properties of the material constituents contained, while the disparity in their physical properties will result in poor uniformity of distribution of reinforced particles. On the contrary, a good uniformity can be obtained by a high-energy ball-milling process, when the ball-powder ratio, the rotation speed and the ball-milling time are well controlled. As a result, the disadvantages caused by the physical property disparities of composite constituents can be avoided. Additionally, the high-energy ball-milling process

can also help to achieve uniform distribution of the submicron reinforced particles. As shown in Figure 2, the results of mechanical mixing and high-energy ball-milling on distributional uniformity are compared, and the latter is obviously superior to the former in particle distributional uniformity.

5 Furthermore, smaller particle size leads to better uniformity. The appropriate ball-powder ratio, the suitable rotation speed and the right ball-milling time are the key factors for obtaining uniform distribution of reinforced particles in the matrix.

10 The interfacial bonding strength between the matrix and the reinforced particles is an important factor affecting the composite property. Forming a high interfacial bonding strength is a crucial stage in producing sound composites. Under the conventional mechanical-mixing powder metallurgy process, the existence of an oxide layer on the Al-base alloy powder is
15 harmful to the bonding strength. The high-energy ball-milling technique adopted in this invention overcomes the flaws in the aforementioned process, laying a solid foundation for a well-controlled and well-bonded interface.

In the invented particulate reinforced Al-based composite, the reinforced
20 particle is uniformly distributed in the Al-based alloy matrix. During high-energy ball-milling, the Al-base alloy powders are formed through steel ball grinding and impact. Meanwhile, the brittle reinforced particles are crushed and compressed with the deformed Al powder and form a cold welded layer on the steel ball surface. Due to continual work hardening, the cold-welding
25 layer formed on the steel ball surface will fall off from the balls and crushed and cold-welded again. Through this repeating process, the fine reinforced particle is mechanically embedded and dispersively distributed into the Al-alloy powders. Figure 3 shows the cold-welding stripes in the composite powders. Deformation of the Al-based alloy powder, the appearance of cold-
30 welding, and relative uniform distribution of the reinforced particle can be seen.

At Step 4, the density of the green compacted billet is about 70 - 80% of its theoretical value, to ensure the linkage of the air-gaps between powders for the next vacuum degassing.

- 5 At Step 5, either using vacuum sintering or vacuum hot-pressing, vacuumizing and heating are simultaneously carried out, finally heating at 450 - 600°C (the specific heating temperature depends on the specific types of matrix powder) under a vacuum of 10^{-2} Pa. Vacuum degassing is used to remove residual gas of the green billet, and the adsorbent water or chemical crystal water and
10 other volatile substances attached on the powders.

The high cost of composite component fabrication is another key factor limiting their applications. Traditional component forming processes, such as hot extrusion, mold forging and machining can still be applicable for the
15 particulate reinforced Al-base composite. However, the cost is very high as compared with Al conventional alloys due to the composite's poor plasticity and machinability, thus limiting the composite's applications. If the composite billet is fully machined into a complex shape component, the cost is extremely high because of machining tooling easily wearing off, much longer machining
20 time required and expensive composite material machined off. Traditional near net shape approaches, such as forging or hot extrusion are not suitable for the composite component forming, due to their poor plastic deformation nature. Step 6 adopts a semisolid near net shape forming technique to fabricate complex-shaped particulate reinforced Al-based composite
25 components. Taking advantage of the thixotropy characteristics of the composite billets and applying a double heating procedure, when the co-existence of solid and liquid phases is obtained, the compact billet can be easily semisolid squeeze casted. This near net shape semi-solid forming process significantly increases the yield of composite material used for
30 fabrication of composite components. Meanwhile, much less machining is required. Figure 4 shows a vacuum hot-pressed composite billet: 35vol%AlNp / 6061Al and 35vol%SiCp / 6061Al. Figure 5 shows the microcosmic structure of a thixotropic composite.

The advantages of this invented particulate reinforced Al-base composite and its component forming process include the following:

- 5 1. The reinforced particles uniformly distribute in the matrix with a good
interfacial bonding in the matrix, ensuring superior mechanical
properties in terms of high strength and high stiffness. Table 1 displays
the properties of several high-performance composites. In addition, by
controlling the reinforced particle size range, good machinability can be
10 obtained.
- 15 2. The invented composite billet fabrication is a simple process and the
ball-milling time is significantly reduced, resulting in a short production
cycle. In the ball-milling process, no surfactant is added, avoiding
deterioration of the material properties. The adoption of a variable-
speed high-energy ball-milling technique effectively avoids the severe
powder cold-welding. Only a small amount of surfactant is added in the
ball milling process. At a given temperature range, composite powder
crushing is accelerated by vaporization of the surfactants and a
20 composite powder with an appropriate particle size is formed.
- 25 3. The invented composite billet fabricating process easily improves the
average size and surface chemical condition of the reinforced particles
as well as their uniform distribution in the matrix. It reduces the
negative effects resulting from the physical property disparity of the raw
material constituents, to form good interface bonding between the
reinforced particles and the matrix. A well-controlled average particle
size and uniform distribution of particles, especially the uniform
distribution of submicron particles in the Al matrix, can be obtained by
30 this process.
4. The semisolid forming technology is invented for making the near net
shape composite components, hence greatly increasing the yield of

composite billets, reducing the machining time, resulting in an overall cost reduction of composite components

5. This invention organically combines the high-energy ball-milling powder metallurgy technology having a capability of producing high-performance composite billets, with the semisolid near net shape forming technology of components. It takes full advantage of the high-energy ball-milling technique with uniform distribution of small-sized particles and sound interfacial bonding with the matrix, thus ensuring that the composite billets having high performance and good machinability. The semisolid near net shape forming process described in the present invention is a more cost effective way for fabricating complex components. As described above, by adopting the two new processes i.e. the high energy balling /milling and the semisolid process, high performance and cost effectiveness of particulate reinforced Al-base composite components can be obtained.

Table 1: Properties of High-performance Composites

Name of composites	Tensile Strength (MPa)	Yield strength (MPa)	Elastic Modulus (Gpa)	Elongation (%)	Fracture reduction area (%)
17vol%B ₄ Cp/6061Al (T6)	470	415	108	2	--
15vol%SiCp/2024Al (T6)	513	453	100	--	3.3
35vol% AlNp/6061Al (R)	495	--	--	--	--

Among of the three composites, the manufacturing method of Composite
5 17vol%B₄Cp/6061Al (T6) is demonstrated in Sample case 1.

SiC reinforced particles and 2024 Al matrix are used in composite
15vol%SiCp/2024Al (T6). The volume percentage of SiC particles is 15%.
Composite 35vol% AlNp/6061Al (R) consists of AlN reinforced particles and
10 6061Al Aluminum alloy. The volume percentage of AlN particles is 35%.
These composites are all made by the present invented processes.

Sample cases

15 To further illustrate this invention and for a better understanding of the
invented products, its fabricating process involved and advantages, several
sample cases are shown below.

Sample case 1

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A composite B₄Cp/6061Al consists of the reinforced B₄C particle with an
average size of 0.92μm, volume percentage of 17%, and the reinforced
particles are uniformly distributed in the Al-alloy matrix.

The production method is: 1) selection of a volume percentage of 17% of the B₄C particles (weight % : 18.1%) ; 2) weigh 543 grams of B₄C powder of 0.92μm, and 2457 grams of 6061Al powder of 105μm, respectively, and 50 kilograms of 6mm high-carbon steel balls, then load all of them into a balling drum and seal the charging door tightly; 3) vacuumize to 5×10^{-1} Pa, then fill in an inert gas nitrogen at the pressure of 1.02×10^5 Pa; 4) ball milling at 15~25°C under water cooling, the balling drum undertakes a 0.5-hour high-energy ball-milling at an initial rotation speed of 125 rpm; then proceeds for a 2-hour ball-milling at an escalated rotation speed of 192rpm; 5) at the end of the ball-milling, add 20ml methanol, vacuumize to 5×10^{-1} Pa, then fill nitrogen gas until the pressure reaches 1.02×10^5 Pa; 6) under the condition of no water cooling, ball mixing for 0.5-hour in the high-energy ball-milling stage at 15 - 80°C with a rotation speed of 125rpm; 7) discharge the powders when ball-milling is finished. The average particle size of the produced composite powder is 70.6μm, with uniformly distribution of B₄C particle of 0.92 μm 8) feed and seal the composite powder in a 120mm×300mm vacuum rubber package, place it in the hydro-cylinder, and subject it to cold isostatic pressing under a pressure of 200MPa holding for 3 minutes. The density of the green billet is 75% of its theoretical density; 9) the green billet is further hot compacted under a vacuum (3×10^{-2} Pa) hot-pressing under 42 Mpa at 550°C, 10) finally, load the compact billet into a specially-designed induction furnace and double heat to 650°C, when reaching a 60 - 70% liquid-phase content, carry out the semisolid squeeze casting. The properties of this near net shape composite billet are shown in Table 1.

Sample case 2

The composite of SiCp/2024Al consists of a volume percentage of 15% of SiC particle of 3.5 μm in diameter .

The production method is as follows: 1) a 15% volume of the SiC particle (wt% of 17.3) and 2024 Al matrix powders are prepared, 2) weigh 519 grams

of SiC powder of 3.5 μ m, 2481 grams of 2024Al powder of 75 μ m, and 40 kilograms of 6mm high-carbon steel balls, respectively, then load them all into a balling drum and seal the charging door tightly; 3) vacuumize to 5×10^{-1} Pa, then fill in nitrogen gas, until a pressure of 1.02×10^5 Pa is reached; 4) under
5 water cooling at 15~25°C, high-energy ball-milling for 0.5-hour at an initial rotation speed of 125 rpm; then proceed with a 2-hour ball-milling at a higher speed of 192rpm; 5) at the end of the ball-milling, add 10ml methanol, vacuumize to 5×10^{-1} Pa, then fill with nitrogen, until the pressure reaches 1.02×10^5 Pa; 6) under a condition without water cooling, high-energy ball-
10 milling at a rotational speed of 125rpm for 0.5 hour at a temperature range of 15 - 80°C ; 7) discharge the powders when the ball-milling terminates. The average particle size of the composite powder is 35 μ m, and the SiC particles (3.5 μ m in average diameter) uniformly distribute in the composite powder; 8) feed and seal the composite powder in a 120mm×250mm vacuum rubber
15 package, followed by placing it into a hydro-cylinder, then cold isostatic pressing at a pressure of 200MPa, holding for 3 minutes. The density of the green billet is 74% of the theoretical density; 9) the green billet is further hot compacted in a vacuum (3×10^{-2} Pa) hot-press at 510°C under a 42Mpa pressure; 10) finally, load the compacted billet into a specially-designed
20 induction furnace and carry out double heating to 610°C, when reaching a 60 - 70% liquid-phase content, proceed to semisolid squeeze casting.

Sample case 3

25 SiCp/6061Al consists of uniformly distributed 35 vol % SiC particles (3.5 μ m) and 6061 Al matrix powder.

The production method: 1) a 35 vol. % of SiC particles (31.2 wt %) and the 6061Al matrix are prepared; 2) weigh 1248 grams of SiC powder of 3.5 μ m,
30 2752 grams of 6061Al powder of 105 μ m, and 40 kilograms of 6mm high-carbon steel balls, respectively, then load all of them into a balling drum; 3) vacuumize to 5×10^{-1} Pa, then fill with nitrogen gas , until reaching a pressure of 1.02×10^5 Pa; 4) under water cooling at 15~25°C , high-energy ball-milling for

1-hour at an initial rotation speed of 125 rpm; then proceeds 4-hour ball-milling at a high rotation speed of 192rpm; 5) at the end of the ball-milling, add 10ml of methanol, vacuumize to 5×10^{-1} Pa, then fill with nitrogen, until the pressure reaches 1.02×10^5 Pa; 6) under the condition of no water cooling, 5 high-energy ball-milling in a temperature range of 15 - 80°C with a rotation speed of 125rpm; 7) discharge the powders when the ball-milling terminates; 8) feed and seal the composite powders in a 120mm×250mm vacuum rubber package, place it in a hydro-cylinder and subject it to cold isostatic pressing at 500MPa , holding for 3 minutes. The density of the green billet is 70% of its 10 theoretical density; 9) the green billet is then further hot compacted by vacuum (3×10^{-2} Pa) hot-pressing at 550°C under a pressure of 42MPa; 10) load the compacted billets into a specially-designed induction furnace and carry out double heating to 660°C, when a 60 - 70% liquid-phase content is obtained, then proceed to semisolid squeeze casting.

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The invention described herein is susceptible to variations, modifications and/or additions other than those specifically described and it is to be understood that the invention includes all such variations, modifications and/or additions which fall within the spirit and scope of the above description.